

Influence of Gas Environment on Synthesis of Silicon Carbide through Reaction between Silicon and Amorphous Carbon in a Solar Furnace at P.S.A. (Plataforma Solar de Almería)

アルメリア太陽エネルギー実験施設 (P.S.A.) の太陽炉を使ったケイ素と無定形炭素からの炭化ケイ素 (SiC) 合成反応への気相雰囲気の影響

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Silicon carbide (SiC) was synthesized from compacted pellet composed of Si powders and amorphous carbon powders through heating in solar furnace under controlled atmosphere (Ar or N₂). Under irradiation of the solar energy flux 1350 kW/m² (ca. 1650°C in terms of measured temperature) for 30 min, Si was completely converted to SiC in Ar atmosphere while the conversion was incomplete in N₂.

[Received March 5, 1998; Accepted May 19, 1998]

Key-words : Amorphous carbon, Silicon carbide, Solar furnace

1. Introduction

Carbon (C) might exist in variety of allotropic forms like graphite, diamond and amorphous carbon. Graphite is in the reference state of C and activity $a(\text{C})$ of graphite is 1. Activity $a(\text{C})$ of other allotropic forms of carbon, like amorphous carbon, is greater than that of graphite and accordingly, by employment of amorphous carbon as a medium for carburising, carbon content higher than the one achieved by carburizing through reaction with graphite might be realized for synthesis of metal carbide MC_x with certain extent of non-stoichiometric homogeneity range or for synthesis of carbonitride $\text{MC}_x\text{N}_{1-x}$.⁽¹⁻⁵⁾ Besides this equilibrium merit of amorphous carbon yielding carbide or carbonitride with comparatively high C content, there is also kinetic merit of using amorphous carbon as a carburization medium. That is, the carburization reaction with active carbon completes faster than that with graphite.

However, active carbon tends to be graphitized when exposed to temperature T higher than 1500°C for prolonged duration and thence, if the graphitization process of the amorphous carbon is completed before the completion of the carburization reaction, equilibrium merit of amorphous carbon to produce carbide with higher C content would not be gained. Thus, it is an advantage to use a furnace with very quick heating capacity and with massive heat capacitance in order to gain full equilibrium and kinetic merits of carburization reaction with amorphous carbon.

Solar furnace is one of such means of heating. In this report, preliminary results obtained from reaction between Si and amorphous carbon in a solar furnace (SF) in Plataforma Solar de América (P.S.A.) are summarized.

2. Experimental

SF (solar furnace) in PSA (Plataforma Solar de América) is capable of intensifying the solar light flux by factor 15000 with the parabolic concentrator composed of segmented concave mirrors. In the present work, 5-cm diameter spot of the peak flux 1350 kW/m^2 was gained from the natural solar irradiance 800 W/m^2 .

Figure 1 depicts schematic constitution of the reaction chamber set in the SF. This chamber is connected to evacua-

tion system and to gas feed line and the reaction in this chamber might be carried out in either vacuum or other arbitrary gas atmosphere. We call this reaction chamber Mini Vac.

Starting material was compacted powder mixture composed of Si (>99.9% pure, -100 mesh; Ventron, Karlsruhe, Germany) and amorphous carbon (Aktivkohle (50 μm) of E. Merck AG, Darmstadt, Germany) cold-pressed with pressure around 400 MPa into a pellet of diameter around 1 cm and height 0.5 cm. Proportion of amorphous carbon was set to be 1.5 times the molar ratio of Si to ensure that the prepared SiC was in equilibrium with the residual free carbon. Unlike mixture of graphite with Si or other metals, mixture of amorphous carbon and Si did not consolidate readily in as-mixed condition and thence, prior to the compacting, liquefied paraffin heated above its melting point was desired to be added to the powder mixture as the binder. It was ensured experimentally¹⁻³⁾ that

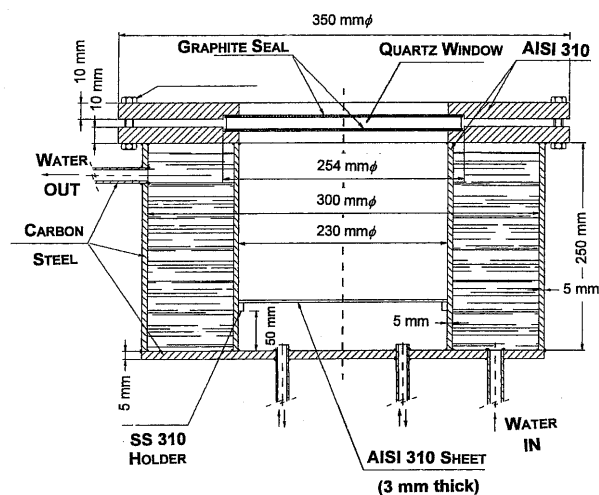


Fig. 1. Schematic cross section of the reaction chamber (MiniVac) for the solar furnace.

paraffin does not exert any noticeable influence on the reaction between carbon and metal because it evaporates before the reaction temperature between the carbon and the metal is reached.

The compacted pellet of the powder mixture composed of Si, amorphous carbon and paraffin held in a graphite crucible was placed in the reaction chamber as depicted in Fig. 1 and exposed to flux of concentrated solar light in the focused spot of the SF. Atmosphere in the chamber was either 99.9% pure Ar (specimen A) or 99.8% pure N₂ (specimen B). In both cases, the maximum temperature (1620°C for A and 1650°C for B) was reached within 3 min and the specimen was kept at the maximum temperature for 30 min. Thereafter, the shutter of the SF was closed and the specimen was cooled in the furnace. It took 5 min to close the shutter completely.

3. Results and discussion

The X-ray diffraction (XRD) patterns for the reaction products are summarized in Fig. 2 and identification of

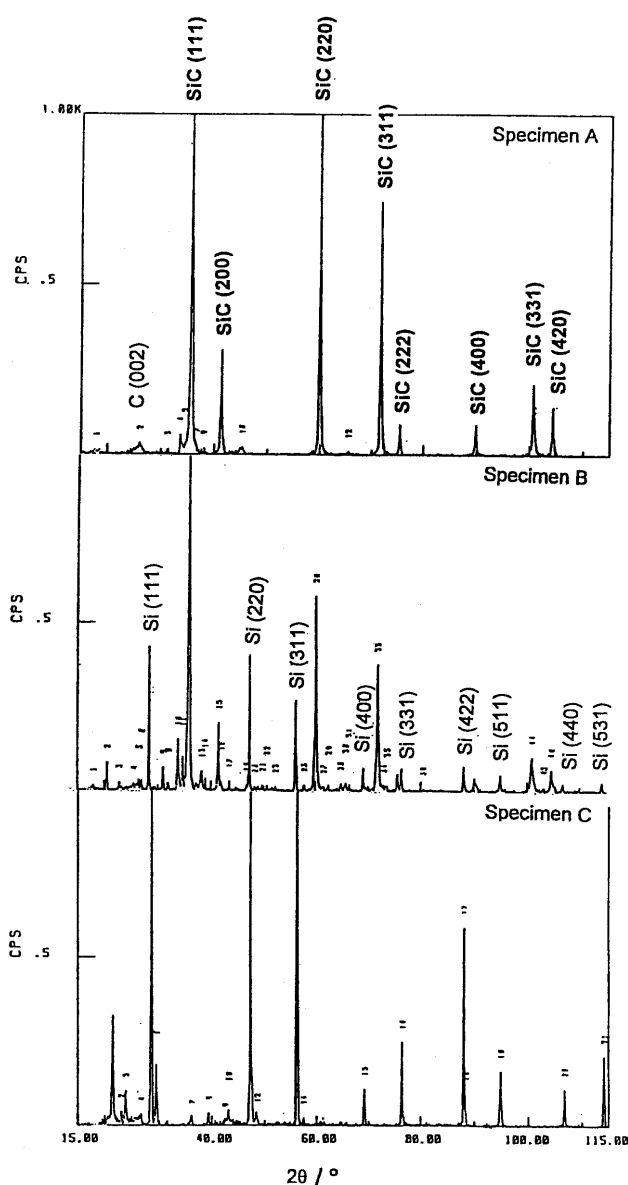


Fig. 2. X-ray diffraction analysis results obtained with Cu K α radiation (45 kV, 20 mA; scan. speed 2°/min).

Table 1. X-ray Diffraction Analysis Results Obtained with Cu K α Radiation (45 kV, 20 mA; Scan. Speed 2°/min)

Identification	Specimen A		Specimen B		Specimen C	
	2 θ / °	Intensity	2 θ / °	Intensity	2 θ / °	Intensity
C (002)	26.20	34	26.60	34	26.75	35
Si (111)			28.50	430	28.57	2749
SiC (111)	35.71	2868	35.74	1727		
SiC (200)	41.44	310	41.49	203		
Si (220)			47.35	405	47.42	1346
Si (311)			56.18	270	56.25	1068
SiC (220)	60.04	1072	60.12	582		
Si (400)			69.21	70	69.25	111
SiC (311)	71.82	749	71.89	377		
SiC (222)	75.54	93	75.66	53		
Si (331)			76.44	70	76.48	252
Si (422)			88.08	76	88.17	591
SiC (400)	90.00	91	90.48	22		
Si (511)			95.05	51	95.04	165
SiC (331)	100.79	212	100.97	103		
SiC (420)	104.47	144	104.64	66		
Si (440)			106.79	26	106.79	111
Si (531)			114.16	30	114.20	210

Intensity: as-counted in cps (counts per second) unit

C(002) peak was broad for any specimen examined

Specimen A: 1620°C for 30 min in Ar at 1.5 bar

Specimen B: 1650°C for 30 min in N₂ at 2 bar

Specimen C: unreacted reference

peaks for these specimens are summarized in Table 1 (specimen C: the unreacted reference specimen composed of Si, amorphous carbon and paraffin). Line identification for Si was made with reference to JCPDS (Joint Committee on Powder Diffraction Standards) diffraction data card 27-1402 (cubic *Fd3m*) and that for SiC with reference to JCPDS 29-1129 (cubic *F43m*; polytype 3C). The results summarized in Table 1 show that Si in the specimen A reacted in inert Ar gas was completely converted to SiC while Si in the specimen B reacted in N₂ gas was not completely converted to SiC. Probable explanation for the retarded SiC formation process in N₂ environment (sample B) might be ascribable to the driving force to form silicon nitride Si₃N₄ in N₂ environment which competed against the SiC formation reaction. In fact, it is well known (e.g., Ref. 6) that high-temperature oxidation behaviour of metallic materials in pure O₂ (or O₂ diluted with inert Ar) is distinguishable from that of the same material in N₂-containing environment (air, for example).

Another important aspect noticed in the obtained XRD patterns is that C(002) peak remained broad after the reaction. This evidence implied that the graphitization of the amorphous carbon did not proceed during the reaction. Peak position of C(002) of graphite is $2\theta = 26.50^\circ$ but C(002) peak position and breadth might vary depending on the degree of deviation of crystallinity of the non-graphitic carbon from that of graphite.^{7,8)}

4. Conclusions

Following conclusions were drawn from the present carbide synthesis experiment using the solar furnace at PSA.

(1) Solar furnace is a useful material processing apparatus allowing very rapid heating (a few minutes to raise the temperature up to around 1600°C from room temperature) for massive amount of specimen material (homogeneous temperature spot diameter about 5 cm). Noting intense energy flux of concentrated solar beam (peak flux as high as 1350 kW/m² from natural solar irradiance 800 W/m²), instantaneous homogeneous heating might be realized for quite thick specimen which is not attainable with other laboratory furnace like IR (infra-red) gold image furnace.

(2) Due to very rapid heating rate to considerably high temperature achievable in the solar furnace, preparation of

carbide with comparatively high C content using amorphous carbon in place of graphite might be undertaken in the solar furnace without problem of graphitization of the amorphous carbon. Spectrum of solar beam did not promote the graphitization of amorphous carbon either in spite of our concern whether it might happen due to certain photo-chemical reaction.

(3) If the metal to be carburized is also a strong nitride former, carbide preparation reaction should be undertaken in inert Ar atmosphere rather than in N₂ gas.

Acknowledgements This work is financially supported by "Training and Mobility of Researchers" Programme of CEC-DG XII (Commission of the European Communities, Directorate General XII for Science, Research and Development). The authors would like to thank Miss Isabel M. A. Duarte, graduate research assistant at INETI (Instituto Nacional de Engenharia e Tecnologia Industrial), for her assistance in specimen preparation and Mrs. Teresa Magalhães at INETI for her help in XRD analysis.

References

- 1) M. Katsura and T. Nomura, *J. Nucl. Mater.*, **51**, 63–68 (1974).
- 2) M. Katsura, N. Shohoji, T. Yato, T. Nomura and T. Sano, "Thermodynamics of Nuclear Materials 1974," IAEA (International Atomic Energy Agency), Vol. II, Vienna (1975) pp. 347–54.
- 3) T. Marcelo and N. Shohoji, *Mater. Chem. Phys.*, **15**, 61–74 (1986).
- 4) N. Shohoji, *Mater. Sci. Technol.*, **3**, 404–10 (1987).
- 5) N. Shohoji, T. Marcelo and M. Katsura, *Solid State Ionics: Diffusions and Reactions*, **38**, 187–94 (1990).
- 6) H. Umehara and N. Koshizaki, *Zairyo-to-Kankyo (Corrosion Engineering; Jpn Soc. Corr. Eng.)*, **46**, 449–55 (1997).
- 7) M. Katsura, T. Yuki, T. Sano and Y. Sasaki, *J. Nucl. Mater.*, **39**, 125–32 (1971).
- 8) T. Nomura, M. Katsura and T. Sano, *J. Nucl. Mater.*, **47**, 58–64 (1971).